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The Liquid Phase Deposition of ZnPtBNs: Study on Structural, Morphology, and Their Sheet-Resistant

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Energy Dispersive X-Ray; ZnPtBNs; Four Point Probe; Sheet Resistance.

Abstract

The ZnPt bimetallic nanoparticles (ZnPtBNs) were successfully prepared through the liquid phase deposition (LPD) method onto the indium-titanium oxide (ITO) substrates at various $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration. The Effect of growth solution, the morphology, structural, and sheet resistance were studied by using field emission electron microscopy (FESEM), energy dispersive X-ray (EDX), X-ray diffraction (XRD) and Four Point Probe (FPP) measurement by using Keithley 2401 source-meter. By inserting a growth solution into a synthesis bottle which is containing ITO substrate and a magnetic stirrer, the ZnPtBNs was successfully in-situ prepared on the substrate for 6 hours at 40°C and stirring speed at 400 rpm. The synthesized ZnPtBNs exhibited homogeneous, fibrous at the (111) orientation with an average diameter in the range of 100-700 nm. The atomic ratio of Zn:Pt and sheet resistance of ZnPtBNs decreased with the increase of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration. The optimal elemental composition of the sample prepared at a ratio of Zn:Pt (1:25) obtained at growth solution of 0.467 mM of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ showed the smallest sheet resistance (13.41Ω) which was 38% lower than the ITO sheet resistance (18.44Ω).

Introduction

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The microstructure, unique-morphology, and elemental composition of a counter electrode (CE) in the dye-sensitized solar cells (DSSCs) devices determine the charge-transferred, electrical conductivity, optical properties and the ohmic sheet resistance (Aberle, Wenham, & Green, 1993; Marjoni Imamora, 2014; Sulistiadji & Pitoyo, 2009; Suryono, 2012). Recently, the research interest has shifted focus on platinum (Pt) as CE (Naumar et al., 2013; Sulistiadji & Pitoyo, 2009; Suryono, 2012; Tang et al., 2012; Umar, Yap, Awang, Salleh, & Yahaya, 2013) due to its high electrical conductivity, chemical stability, obtained quickly, and availability (Umar & Yap, 2013). However, an effort to further enhance the properties through a simple, cheaper, and friendly method to environmental still remains as an attractive issue, especially in the field of nanotechnology (Amrullah, 2016; Y.-L. Lee et al., 2010; Sulistiadji & Pitoyo, 2009; Umar, Yap, Awang, & Salleh, 2017).

An introduction of another metallic material into Pt to form bimetallic nanoparticles has been extensive used in the previous report. They include AgPt (Mulyadi, 2010), CuPt (Umar et al., 2017), AuPt, PdPt (Mulyasa, 2006), etc. The Pt nanoparticles in the bimetallic form have unique properties such as excellent optical, electronic, and electrical (Ali Oemar, 2008; Kurnia, Imamora, & Maiyena, 2019; Sabri, 2010). Besides, the Zn as a metallic material that has good electrical conductivity, excellent availability, anti-corrosion, and low-cost materials could enhance the initial conductivity of the Pt, an at the same time introduce new

bimetallic elements as a counter electrode in DSSC devices (Barus, Syahrin, Arifin, & Hamdan, 2015; Y.-L. Lee et al., 2010; Marjoni Imamora, 2014; Yin, Zhang, & Zeng, 2008).

There are several well-known methods to synthesize Pt nanoparticles, namely reduction of metal precursors, dry plasma reduction, electrochemical deposition, and direct growth of metal on substrates by using the liquid-phase deposition method (LPD) (Chang, Oyama, & Hirao, 2006; Farasdaq; Li et al., 2008; Moraes, Saito, Leite, Massi, & da Silva Sobrinho, 2016; Mulyasa, 2006; Umar et al., 2017; Yoon, Vittal, Lee, Chae, & Kim, 2008). Among these methods, LPD is often used to produce Pt nanoparticles (Kharisov, 2008; Sulistiadji & Pitoyo, 2009; Umar et al., 2017).

This paper reports preparation of the fibrous ZnPt bimetallic nanoparticles (ZnPtBNs) by controlling the concentration of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, K_2PtCl_6 , SDS, and formic acid by using LPD method. The morphology of resultaning samples was observed using FESEM. The shifts on the diffraction peaks were detected by X-ray diffraction (XRD) spectra to confirm the formation of ZnPtBNs. Meanwhile, the elemental composition of the sample was obtained using the Energy Dispersive X-ray (EDX).

Methodology

Preparation of ZnPtBNs

The ZnPtBNs were prepared using the LPD method in deionized (DI) water. In the first process, clean ITO was put into growing solution (15 ml) containing 1 mM of potassium hexaachloropaltinate (IV) (K_2PtCl_6), 0.01 M sodium Dodecyl sulfate, 1 mM formic acid and zinc nitrate Hexahydrate ($\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$) at five different concentrations (0.066, 0.200, 0.333, 0.467 and 0.660 mM). All chemical were purchased from Sigma Aldrich. The growth solution was stirred at 400 rpm for 6 h at 40°C. The ZnPtBNs was obtained on the surface of the ITO substrate. Next, ZnPtBNs was taken and rinsed with DI water and dried at temperature of 70°C for 10 minutes.

Characterization of ZnPtBNs

The morphology of ZnPtBNs was characterized by using the FESEM (Zeiss Supra 55VP) model at an acceleration voltage of 30 kV. Furthermore, the elemental composition of the ZnPtBNs on the ITO substrate was obtained using EDX. The X-Ray Diffraction (XRD) and FPP measurements were done using Keithley 2401 source meter, respectively.

Results And Discussion

Morphology of ZnPtBNs

The fibrous ZnPtBNs were successfully synthesized on the ITO substrate. The FESEM images (see Fig. 1A-1E) show the bimetallic nanoparticles efficiently covered the entire ITO substrates. The insets on respective of Figure 1A-1E show the change of ZnPtBNs morphology from cubic (inset of Fig. 1A), spherical (inset of Fig. 1B and 1C), and flowers (inset of Fig. 1D and Fig. 1E) structure with the particles size of 40-700 nm (see Fig. 1F). The Pt nanoparticles usually have morphology of cubic (H. Lee, Habas, Somorjai, & Yang, 2008; Sulistiadji & Pitoyo, 2009). They slightly changed to spherical (Kulesza & Faulkner, 1989), and flower-like form due to the increase of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration into the Pt as a host growth solution (Tang et al., 2012). The ZnPtBNs morphology also changed the miscibility, leading the formation of new morphologies that are rough and fibrous. Besides, the obtained microstructure in the flower-like morphology was useful to enlarge an active surface area, making them more attractive as an excellent electrical conductivity material (Kharisov, 2008; Tang et al., 2012). However, the flower shape becomes irregular and tends to damage to form smaller particles at addition of high Zn concentrations of >0.660 mM (see Fig. 1F). The crooked occurred as an effect of the saturation of Zn^{2+} attached to the Pt structure. The optimum, and homogeneous of fibrous flower ZnPtBNs morphology was successfully obtained at a concentration of 0.467 mM.

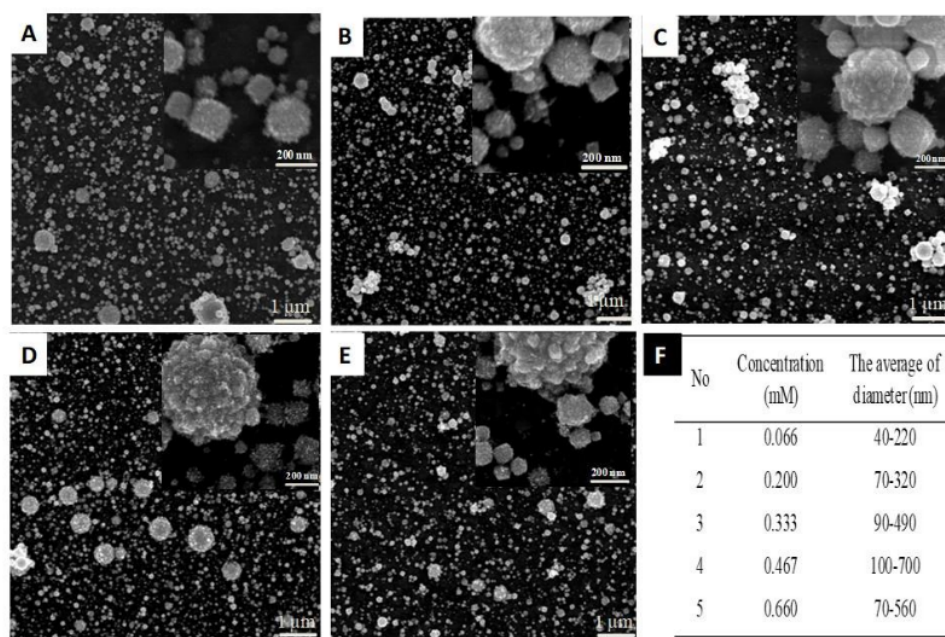


Fig. 1 FESEM images of ZnPtBNs prepared by the addition of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ at (in mM) **A.** 0.066, **B.** 0.200, **C.** 0.333, **D.** 0.467, **E.** 0.660, and **F.** The diameters of ZnPtBNs were obtained at such above concentration. The Insets show the corresponding enlarged FESEM images.

The EDX characterization of a selected ZnPtBNs is shown in Fig. 2A, providing the atoms percentage of the elemental composition information of ZnPtBNs at five different concentrations (see Table 1). Table 1 describes the sample with 15 ml growth solution at a level of Zn had a comparative value with different atomic numbers. It is due to the addition of Zn in nanoparticle ZnPt (Naitabdi et al., 2018). Interestingly, the increase of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ level decreased the ratio Zn to Pt (see Table 1), which was in a good agreement with the FESEM analysis in Fig. 1A-IF. At a high Zn and Pt ratio resulted in more hairy-entities attach to the Pt, promoting the formation of the spherical and fibrous flower of ZnPtBNs, and leading gradual increment of the particles sizes with the $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration increased as well (see Fig. 1F).

Figure 2B shows the XRD spectrum of ZnPtBNs, which were studied using Bruker D8 XRD with $\text{CuK}\alpha$ at a scan rate of $0.025^\circ/0.1$ s. There were three XRD peaks of fibrous ZnPtBNs (JCPDS card number 03-065-3257) detected at diffraction positions of $2\theta=40.08^\circ$, 46.62° , and 68.06° correspond to the crystal planed diffraction of (111), (200), and (220), respectively. These peaks seem similar to Pt diffraction peaks, namely $2\theta=39.76^\circ$, 46.23° , and 67.45° (JCPDS card number 70-2057) associated with (111), (200), and (220) as well. These shifts of the diffraction peaks confirmed the formation of ZnPtBNs. Interestingly, the diffraction intensity of (111) crystal plane is higher than (200) and (220), showing the ZnPtBNs growth was oriented to (111) plane, which was face-centered cubic (fcc) with a lattice of 3.89 Å. Besides, the diffraction intensity of (111) also increased at higher $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentrations. Such finding is a good agreement with the FESEM results shown in Fig. 1A-1E. Besides, the sample also detected another peaks at $2\theta=21.54^\circ$, 30.24° , 35.22° , 50.54° , and 60.03° , which are believed to originate from the ITO substrate.

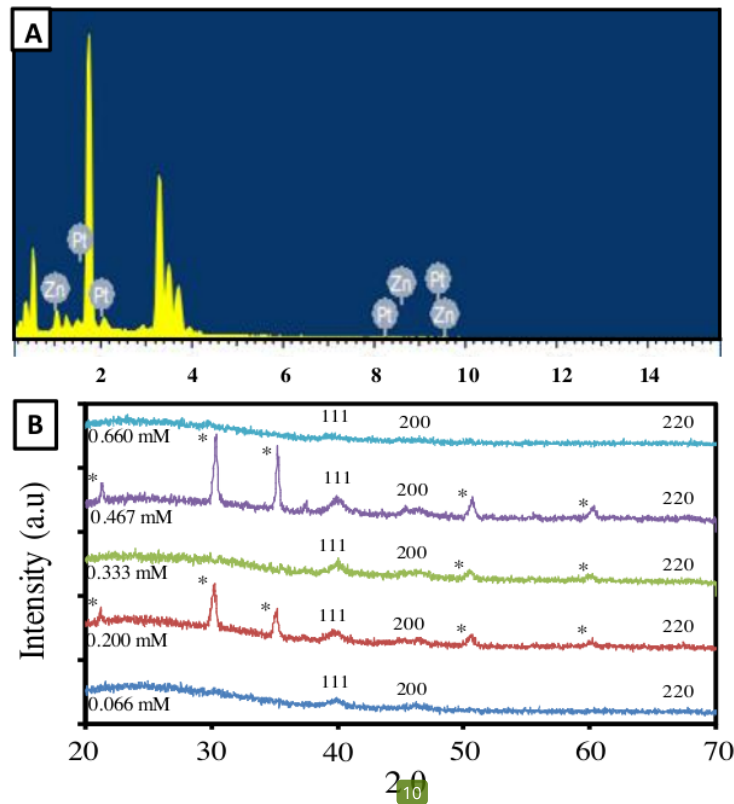


Fig. 2 A) The EDX curve of ZnPtBNs obtained at $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration of 0.467 mM. B) The XRD spectra of ZnPtBNs prepared at differences concentrations of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$.

Table 1 The atomic ratio of Zn:Pt from ZnPtBNs, which were prepared by additions of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ with various concentrations obtained from EDX analysis.

No	Concentration (mM)	Elements of Atom (%)		Atomic Ratio	
		Zn	Pt	Zn	Pt
1	0.066	2.47	97.53	1	40
2	0.200	2.64	97.36	1	37
3	0.333	3.28	96.72	1	30
4	0.467	3.78	96.22	1	25
5	0.660	4.69	95.31	1	20

Fig. 3 shows the sheet resistance and average diameter of ZnPtBNs obtained at various $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ concentration. Based on FPP characterization results, the ZnPtBNs sheet resistance was proportional with the Zn concentration. Since the lower amount of Zn and Pt was an incomplete process at a low level of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, it caused higher amount of hydroxyl (OH^-) molecule leading to a higher sheet resistance, as also reported earlier (Naitabdi et al., 2018). The increase of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ amount resulted in the Zn^{2+} increased and reduced to a Zn^0 state at Pt (111). The electron transfer from $n=2$ to $n=0$ will be induced OH^- adsorption

takes place as OH adsorption onto Zn at Pt(111) during the synthesis process (Igarashi, Aramata, & Taguchi, 2001). Such process is believed to reduce the number of hydroxyls (OH⁻) at higher of Zn(NO₃)₂·6H₂O concentration, resulting in a lower sheet resistance. However, as can be seen in Fig. 3 on the level of 0.200 mM, the sheet resistance of ZnPtBNs is quite low (14.13 Ω), and then increases again to 229.03 Ω at a concentration of 0.333 mM. The instability of the trend represents the change of the structure (see Fig.1A-1E). The stable and the optimum sheet resistance value was obtained at a concentration of 0.467 mM (±13.41 Ω), showing 38% lower than the ITO sheet resistance (18.44 Ω). The finding is such in a good agreement with FESEM analysis, where a concentration of 0.467 mM, the ZnPtBNs microstructure resulting from the fibrous flower form, leading the formation of larger an active surface area (Chen et al., 2011) and good electrical conductivity or lower sheet resistance (Yoon et al., 2008). These results are excellent in line with an average diameter of ZnPtBNs, where it gradually increased with Zn(NO₃)₂·6H₂O concentration, promoting the formation of ZnPtBNs which had a larger surface area at (111) crystal plane (Hsieh, Wei, Hsiao, & Chen, 2012).

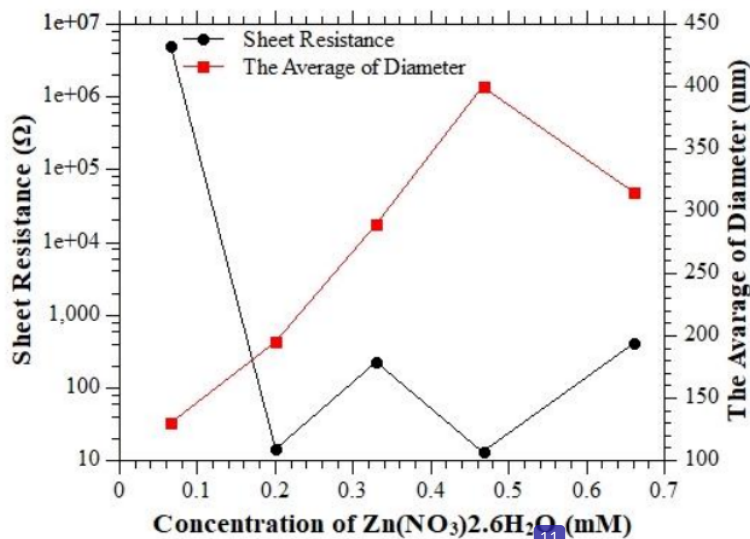


Figure 3 The sheet resistance and the average diameter (nm) as a function of Zn(NO₃)₂·6H₂O concentration.

Conclusions

The Effect of Zn(NO₃)₂·6H₂O concentration addition on morphology, structural, and electrical properties, especially sheet resistance of the fibrous ZnPtBNs prepared using the LPD method, have been successfully investigated. The morphology of fibrous ZnPtBNs was slightly changed from cubic to spherical, and hairy flower at higher due Zn(NO₃)₂·6H₂O concentration

into the growth solution of Pt. The best morphology of the synthesized ZnPtBNs was a flower-like with homogeneous form, more fibrous with particle sizes ranging from 100-700 nm. The EDX and XRD results show the ratio of Zn to Pt decreases, and the diffraction intensity of (111) increases with the addition of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ increased. The best atomic ratio of Zn: Pt of 1:25 and the lowest sheet resistance of $13:41 \Omega$ were obtained at a concentration of $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ at 0.467 mM.

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